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Effects of two grading techniques of zirconia material on the fatigue limit of full-contour 3-unit fixed dental prostheses

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ABSTRACT

Objective. This study evaluated the effects of two grading techniques of zirconia material on the fatigue limit of full-contour 3-unit fixed dental prostheses (FDPs).

Methods. Presintered blocks of 3Y-TZP were milled to obtain sixty-nine 3-unit FDPs, which were divided into three groups (n = 23). The control group (CTL) was sintered and glazed following manufacturer's instructions. In the two experimental groups presintered FDPs received a surface silica/glass infiltration treatment before the sintering process. Silica sol-gel group (SSG) was graded by the sol-gel processing route, while the glass-zirconia-glass group (GZG) was graded by an enameling technique. Graded groups did not receive a glaze layer after sintering. All FDPs were then luted with a dual-curing resin cement on composite abutments, embedded in polyurethane and stored in water for five days. The initial load of the fatigue test was calculated based on the results of the monotonic testing applied on three specimens of each group. To determine the fatigue limit, 20 samples of each group were subjected to staircase testing (100,000 cycles/5 Hz).

Results. The fatigue limits (in Newtons) were CTL = 1607.27, SSG = 1824.31, and GZG = 2006.57, and the Dixon and Mood test indicated statistically significant differences among groups (95% confidence interval) (GZG > SSG > CTL).

Significance. The infiltration of silica and glass on bulk zirconia, by two different grading methods, increased the fatigue limits of monolithic zirconia FDPs.

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1. Introduction

The advances in CAD/CAM (computer-aided design and manufacturing) technology made possible to produce full-contour or monolithic zirconia restorations with occlusal design that do not need to be veneered, thus avoiding the clinical problem of chipping of the veneering porcelain [1–3]. Conservative monolithic zirconia restorations with less than 1 mm thickness can be milled and still present strength similar or superior to that of metal–ceramic and lithium disilicate crowns [4–6]. However, concerns like the challenge of promoting enhanced adhesion between substrate and zirconia ceramic [7], aging (hydrothermal degradation) [8,9], and wear of antagonist teeth [10] have limited the use of monolithic zirconia.

When subjected to fatigue and moisture, zirconia-based materials suffer from stress corrosion, and their strength is compromised [9,11–14]. In the veneered zirconia restoration, the overlay porcelain protects zirconia against direct exposure to moisture, pH and temperature variations, and impact from chewing forces [11]. Conversely, monolithic zirconia restorations would have only a thin glaze layer on top. Indeed, investigators [15] have shown that glaze application prevents hydrothermal degradation. But, we must also consider that such a thin layer may be removed during function (wear) [16] and, consequently, that zirconia would again be prone to aging. In fact, recent research has shown that a glazed zirconia surface causes greater wear of the antagonist tooth relative to an as-polished surface [16,17]. Nevertheless, Hmaidouch and Weigl [2] considered that studies on this topic are subject to both evaluation and confounding bias, making it impossible to associate tooth wear with any specific causal agent.

A promising technique recently proposed for dental applications is the grading of zirconia [18–20]. The glass infiltration method, proposed by Zhang and Kim [20], on the surfaces of 3Y-TZP aims to promote better stress distribution, resulting in enhanced flexural strength compared with that achievable with monolithic zirconia [21]. The glass layer on the external surface also protects against hydrothermal degradation, avoids antagonist wear, and allows for color variations, improving the esthetics of monolithic restorations [22]. At the same time, the glass layer formed on the cementation surface of the restoration makes etching and silanization possible, improving adhesion to resin cements [23].

The main feature of functionally graded materials (FGM) such as the zirconia described above is a gradual variation of composition and/or phases, and consequently properties, inside the same material [24–26]. Another FGM processing route is silica infiltration of zirconia by a sol–gel method, which allows for the modification of the zirconia surface by reactive diffusion of silica into zirconia during the sintering process. This method does not require additional thermal treatment and forms a zirconium silicate, which affects the mechanical properties of zirconia and improves its structural homogeneity [27].

However, the ability of these next-generation functionally graded 3Y-TZP multiple-unit FDPs to resist fatigue damage remains unknown. Therefore, the purpose of this current study was to evaluate the effect of glass/silica infiltration on the fatigue limit of 3-unit fixed dental prostheses manufac-

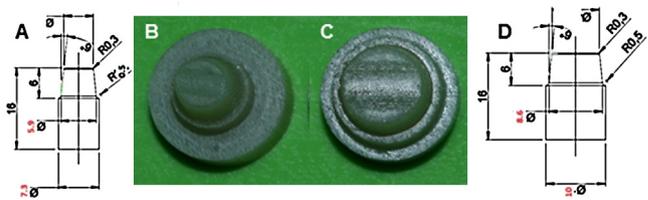


Fig. 1 – Specifications for premolar (A) and molar (D) preparations; (B, C) the dies were milled from dentin composite analogue (NEMA grade G10). Based on Campos et al. [28], 2016.

tured with monolithic 3 mol% Y-TZP. The null hypothesis was that the fatigue limit of monolithic zirconia bridges would not be affected by glass/silica infiltration.

2. Materials and methods

2.1. Materials

Sixty-nine presintered 3Y-TZP blocks (40 × 15 × 14 mm, VITA In-Ceram YZ Vita for inLab, VITA) were milled to obtain 3-unit FDPs extending from the second mandibular premolar to the second mandibular molar. Main composition of the VITA In-Ceram[®] YZ for inLab[®] provided by the manufacturer is: zirconium dioxide (ZrO₂) 92.6 %mol, yttrium oxide (Y₂O₃) 5.3% wt %, hafnium oxide (HfO₂) < 3 wt%, aluminum oxide (Al₂O₃) and silicon dioxide (SiO₂) < 1 wt%.

2.2. Specimen preparation

The dies (2nd mandibular premolar and 2nd mandibular molar) were milled from a dentin composite analogue (NEMA grade G10, Accurate Plastics Inc., Falmouth, MA, USA). The prosthetic preparations had the following characteristics [28]: 6 mm in height; 120° circumferential chamfer (radius = 0.5 mm); 6° occlusal convergence angle (Fig. 1A, D); all transitions from the axial to the occlusal surfaces were rounded, and a central mesiodistal sulcus was simulated on the occlusal surface (Fig. 1B, C).

Using an adapted surveyor for keeping long-axis of preparations perpendicular to the ground (x axis), the abutments were embedded into polyurethane (F 16Polyol e F 16 ISO, Axson Technologies, Eaton Rapids, MI, USA) at distances of 17 mm from each other. The obtained sets were scanned (InEos, Sirona Dental Systems GmbH, Bensheim, Germany), and the corresponding three-unit FDPs were created in a CAD system (inLab SW4.2, Sirona) according to the requirements for full anatomic restorations of posterior teeth: the minimum wall thickness was 0.7 mm in the occlusal area, 0.7 mm from the bottom of the fissure, 0.5 at the lateral walls, and 0.2 mm at the cervical region, and anatomical connectors were within the minimum section area of 9 mm². The milling of the restorations was performed with a CAM unit (CEREC InLab MC XL, Sirona Dental Systems GmbH, Bensheim, Germany), under constant water cooling. After being milled, the restoration was separated from the block-holder, and each one of them was evaluated by stereomicroscopy (Discovery V20, Carl Zeiss,

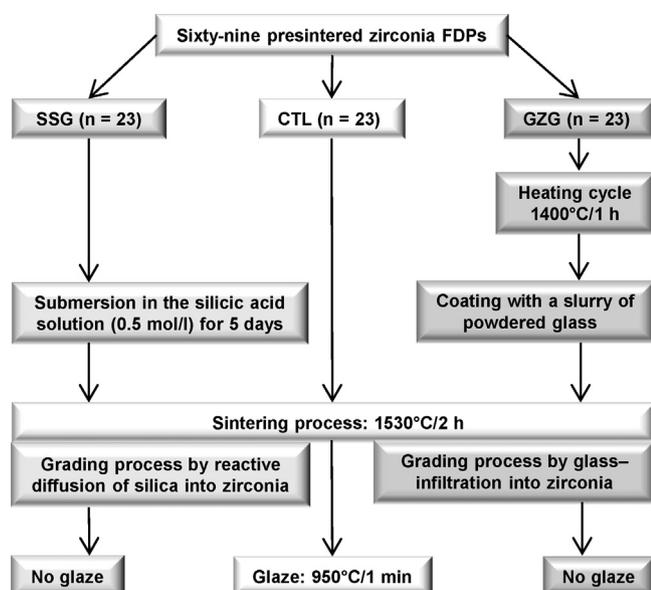


Fig. 2 – The grading and sintering protocols of each group.

Göttingen, Germany) at 7.5 \times and 25 \times magnifications. Grooves on the vestibular wall (close to connector) and mesial walls (where the block holder was cut) of the premolars were seen in all FDPs. The restorations were then cleaned with distilled water (grade 3) in an ultrasonic bath (Digital Dental Ultrasonic CD-4820, Shenzhen Codyson Electrical Ltd., Guangdong, China) for 10 min and left undisturbed to dry in air.

2.3. Grading and sintering procedures

After the milling and cleaning process, the 69 FDPs were divided into three experimental groups ($n=23$), according to the grading protocols (Fig. 2).

2.3.1. Control group (CTL)

FDPs were sintered in an appropriate furnace (Vita Zyrcomat T, VITA) as follows: time for increasing temperature, 1.5 h \rightarrow ending temperature, 1530 $^{\circ}$ C \rightarrow holding time at ending temperature, 2 h \rightarrow cooling to 400 $^{\circ}$ C with firing chamber being closed. After cooling, FDPs were glazed (VITA Akzent Plus Glaze, VITA) accordingly: time for increasing temperature, 11.37 min \rightarrow ending temperature, 950 $^{\circ}$ C \rightarrow holding time at ending temperature, 1 min.

2.3.2. Silica sol–gel group (SSG)

FDPs underwent silica infiltration by the sol–gel method, according to Campos et al. [27]. The silicic acid was obtained by the passing of sodium metasilicate ($\text{Na}_2\text{SiO}_3 \cdot 5\text{H}_2\text{O}$) aqueous solution (10% m/m) through an ion-exchanging resin (IR120—Rohm and Haas). FDPs were submerged in the silicic acid solution, 0.5 mol/l, for 120 h in a hermetically sealed container with the occlusal surface facing down, at room temperature. FDPs were then removed from the gel and allowed to dry at room temperature, with the occlusal surface facing up. They were then sintered according to the parameters described above, in the same furnace. The FDPs did not receive a glaze layer.

2.3.3. Glass-zirconia-glass group (GZG)

The glass infiltration followed the protocol proposed by Zhang and Kim [20]. Presintered FDPs were heated to 1400 $^{\circ}$ C for 1 h in air. After cooling, all accessible surfaces were coated with a slurry of powdered glass composed of SiO_2 (65.5%), Al_2O_3 (11.7%), K_2O (10.0%), Na_2O (7.3%), CaO (3.0%), and Tb_4O_7 (1.9%). The glass composition was formulated so that the glass and zirconia had similar coefficients of thermal expansion (CTE). Glass infiltration and densification were carried out simultaneously at 1530 $^{\circ}$ C for 2 h in air. As aforementioned, the FDPs were not glazed also.

2.4. Luting procedure

The composite dies were treated with hydrofluoric acid (10% CondAC-porcelana, FGM, Joinville, Brazil) for 1 min, rinsed with water, and dried, after which they received one layer of silane agent (RelyXTM Ceramic Primer, 3M ESPE, St. Paul, MN, USA).

The cementation surfaces of 3-unit zirconia bridges were treated according to the recommended protocols for each production method.

2.4.1. CTL group

FDPs were abraded with 30 μm silica-coated alumina particles (Rocatec Soft, 3M ESPE) with pressure of 2.8 bar, distance of 10 mm, and time of 10 s, followed by silanization (RelyXTM Ceramic Primer, 3M ESPE) for 1 min.

2.4.2. SSG group

FDPs were etched with 2% hydrofluoric acid (Fórmula e Ação Farmácia, São Paulo, Brazil) for 15 s. The adhesive (Single Bond Universal, 3M/ESPE) and optimizer agent (OPC, 3M do Brasil Ltda, São José do Rio Preto, SP, Brasil) were mixed (1:1) for 5 s, and rubbed over the cementation surface for 20 s. The excess was removed with an oil-free air jet, and polymerization was performed for 10 s (Radii-cal High Powered LED, SDI, Victoria, Australia).

2.4.3. GZG group

FDPs were etched with 5% hydrofluoric acid (Fórmula e Ação Farmácia, São Paulo, Brazil) for 1 min, rinsed with water, and dried. Adhesive and optimizer agent were applied following the same steps aforementioned.

A dual-cured resin cement (Variolink II, Ivoclar Vivadent, Schaan, Liechtenstein) was poured and mixed as recommended by the manufacturer and applied to the intaglio surfaces of the FDPs, which were seated on the preparations. Each bridge was then held in position by a metal rod (26.45 oz) positioned in the center of the occlusal surface, excess cement was removed, and each abutment was light-activated for 40 s on the buccal, lingual, and occlusal surfaces by a LED curing light with a wavelength ranging from 440 to 480 nm (Radii-cal, SDI). Then, the cemented sets (Fig. 3A) were positioned in a rectangular metallic matrix, which was filled by polyurethane (F 16Polyol and F 16 ISO, Axson Technologies, Eaton Rapids, MI, USA) until 1 mm apart from the finishing line of preparations (Fig. 3B). This made possible an easily insertion of the polyurethane blocks on the device designed for mechanical testing (Fig. 3C). After the polyurethane was cured, the sets

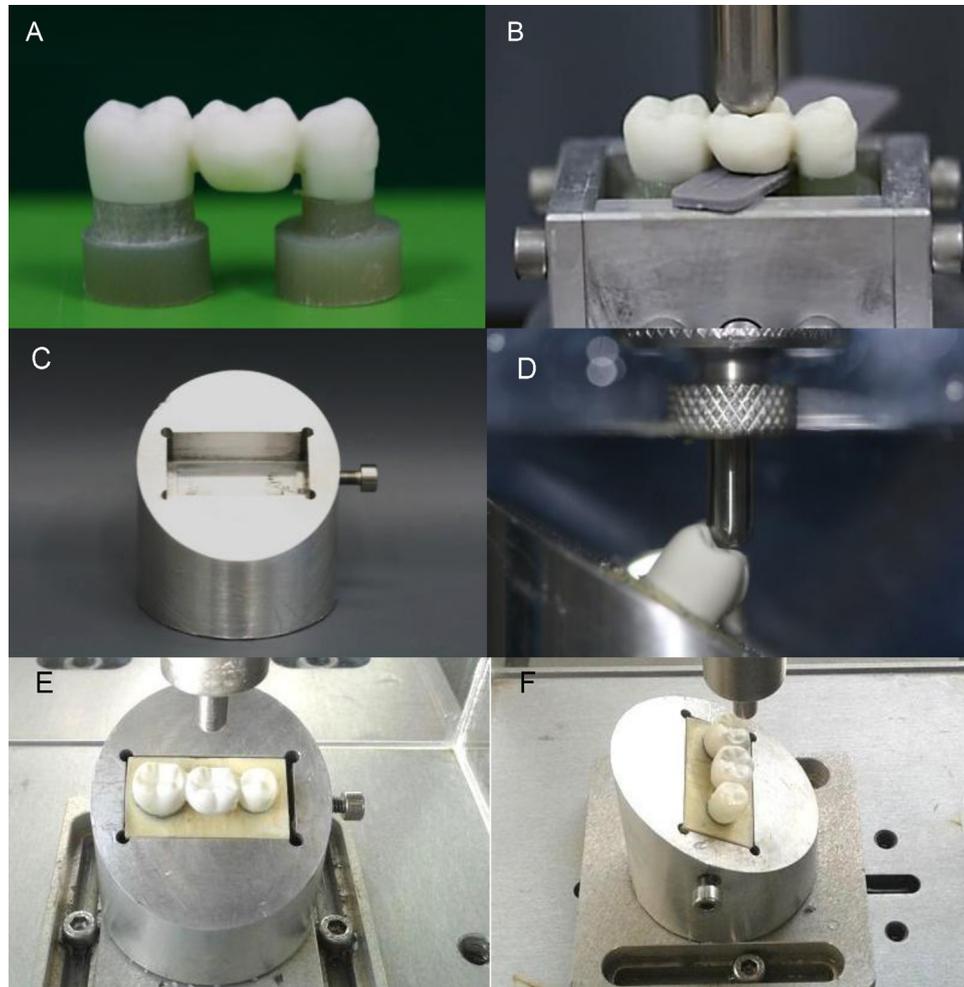


Fig. 3 – (A) FDP luted on the composite preparations; (B) the cemented sets positioned in a rectangular metallic matrix for embedding on polyurethane; (C) aluminum device designed for mechanical testing shows a rectangular site for the correct insertion of the polyurethane blocks; (D) samples tilted 30° relative to the horizontal plane. A tungsten-carbide tip contacts the pontic on its inner inclines of the buccal cusps; (E, F) sample positioned for staircase testing. The mesio-distal (E) and buccal-lingual position (F) relative to the tip.

were stored in grade 3 distilled water for five days before the fatigue test.

2.5. Fatigue limit evaluation (staircase test)

Before fatigue testing, the monotonic failure loads were determined from three specimens of each experimental group. In a universal testing machine (Emic DL 1000, Emic, São José dos Pinhais, Brazil), samples were tilted 30° relative to the horizontal plane, immersed in water, and a tungsten-carbide round tip (6 mm in diameter) applied a crescent load to the inner inclines of the buccal cusps of the pontic (1st molar) until the samples fractured (load cell 1000 Kgf, 1 mm/min) (Fig. 3D).

The starting values of the staircase test were obtained by using 70% of the mean of monotonic load-to-fracture of 3 samples from each experimental group. The purpose of this procedure was to obtain a value near the endurance limit (stress below which failure never occurs, even for an indefinitely large number of loading cycles) of each group. The staircase test was performed in an electric machine (Instron

Electropuls E-3000, Instron, Glenview, IL, USA), with the samples arranged at the same manner that were positioned in the monotonic testing (Fig. 3E, F). The test was set at 100,000 cycles, at 5 Hz frequency. The load increment (or step) was defined as 5% of the initial load. Samples were tested sequentially. Each experimental group followed its own load regimes, because the initial load and step sizes were derived from the respective means of load for fracture. Thus, the first sample was tested under the predetermined initial load for 100,000 cycles. If it survived, it was counted as a runout or 'pass' and its successor was tested one stress step higher. If the sample failed, its successor was tested one step lower. Twenty samples were tested for each group. Upon completion of testing, it was determined the less frequent event (failure or survival), which was used in the analysis. Then, the data were tabulated in five-column table: (I) load values experienced by the less frequent event; (II) load levels are numbered in order of increasing magnitude, where the number 0 is assigned to the lowest load value (i); (III) number of times the event occurred at each load level (ni); (IV) the product of $i \times ni$; (V) the prod-

uct of ($i^2 \times n_i$) and the letters N, A e B (the sum of n_i , $i \times n_i$ e $i^2 \times n_i$, respectively). Those data allowed the determination of the fatigue limit (FL) and the standard deviation (SD) from equations [29]:

The statistical estimate of mean fatigue strength was calculated by:

$$S_m = S_0 + d (A/N \pm 1/2)$$

where S_0 = lowest load value experienced by the less frequent event; d = load increment value; A = sum of the column IV; N = total number of the less frequent event.

The plus sign (+) was used if the more frequent event was survival and the minus sign (–) was used if the more frequent event was failure.

For the standard deviations the equation was:

$$\sigma = 1.62 \times d \left((NB - A^2) / N^2 + 0.029 \right)$$

and for confidence interval:

$$S_m - 1.96 \times \sigma_m \leq \mu \leq S_m + 1.96 \times \sigma_m$$

2.6. Data analysis

The Kolmogorov–Smirnov test was applied to test data distribution and the Grubbs test for testing the presence of outliers at a significance level of 5%. After that, data were subjected to the Dixon and Mood method, which involves maximum-likelihood estimation techniques for analytical solutions to the problem of determining the mean and standard deviation [29,30]. This method assumes that the fatigue limit follows a normal distribution. The two statistical properties are determined by either using only the failures or only the survivals, depending on the least frequent event that had the smaller total numbers. The fixed stress increment should be in the range of half to twice the standard deviation of the fatigue limit.

However, the Dixon and Mood method cannot be used in the following situations:

- The fatigue limit is not normal and cannot be transformed to be normal
- The stress increments are not identical at the difference stress levels
- The stress increment is greater than twice the standard deviation.

In such cases, or when the data follows a Weibull distribution, the statistical analysis should be done by Zhang and Kececioglu method [31,32].

2.7. Failure analysis

Fracture analysis was firstly performed by means of a binocular stereomicroscope (Discovery V20, Carl Zeiss, Göttingen, Germany) for determination of the spatial relationships of the observed fracture features [33,34]. The fragments were then cleaned in an ultrasonic bath (Ultrasonic CD-4820, Shenzhen Codyson Electrical Ltd., Guangdong, China) with isopropyl alcohol for 10 min, dried, and coated by gold-sputtering (Emitech SC7620, Quorum Technologies Ltd., Laughton, UK). They were then analyzed by scanning electron microscopy (INSPECT S50, FEI, Brno, Czech Republic).

3. Results

3.1. Data distribution

The Kolmogorov–Smirnov test indicated that there was no violation of the normality assumption, and Grubbs' test did not detect outliers (at the 5% level of significance).

3.2. Fatigue limit—staircase test

The normal data distribution, the step values ranging from half to twice the standard deviation (fixed for each group), the sample size higher than 15, and three or more load levels for failure and survival allowed for the use of Dixon and Mood data analysis, with low risk of bias [30,31,35,36]. Table 1 presents parameters for staircase test and the results.

Statistical differences were observed among the fatigue limits of the three groups, given that the non-overlapping of the confidence intervals was noted: the control group had the lowest fatigue limit, while GZG group obtained the highest one. Still, the control group presented the lowest decrease regarding initial load to fracture (15.8% decrease), and the GZG group presented a decrease of almost 40% regarding initial load to fracture (Table 1). The SSG group presented the lowest standard deviation.

The profiles generated by the staircase test are presented in Fig. 4.

Table 1 – Mean of single-cycle fracture load (in Newton), parameters [initial load (IL in Newton) and step size (SS in Newton)], and results [fatigue limit (FL in Newton), standard deviation, confidence intervals, and decrease from fracture load to fatigue limit (decrease in %)]. Abbreviations: (CTL) control group; (SSG) silica sol–gel group (GZG); glass-zirconia-glass group.

Groups	Fracture load (N)	Staircase parameters		Staircase results		
		IL (N)	SS (N)	FL (N) ^a	CI (95%)	Decrease (%)
GZG	3235.50	2264.85	113.24	2006.57 (104.2) ^a	1934.3–2078.7	37.9
SSG	2619.17	1833.42	91.67	1824.31 (69.7) ^b	1759.5–1889.1	30.3
CTL	1907.66	1335.36	66.77	1607.27 (91.4) ^c	1542.9–1671.6	15.8

^a The statistically significant differences are indicated by different superscript letters.

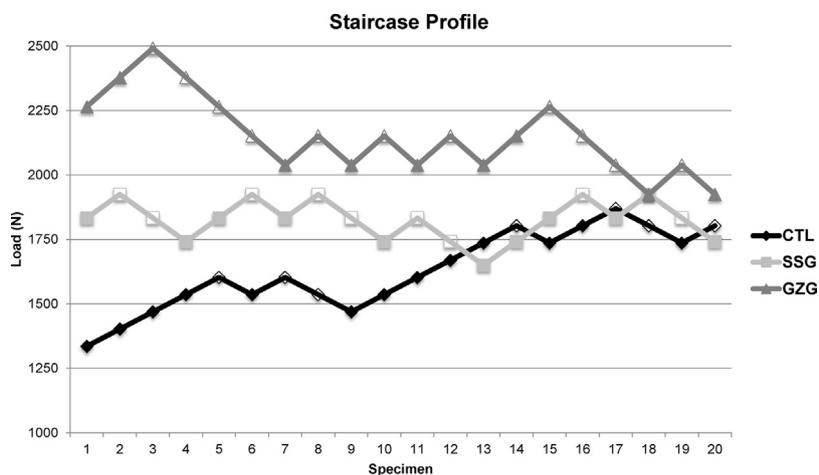


Fig. 4 – Staircase profiles generated after the test for each group. Fractured specimens are represented by empty forms.

	Mesial connector	Occlusal	Distal Connector
CTL	 03 specimens	 02 specimens	 02 specimens
SSG	 07 specimens		 03 specimens
GZG	 02 specimens	 06 specimens	 04 specimens

Fig. 5 – Different failure patterns found for each test group.

3.3. Failure analysis and topographic features

All failures were catastrophic, but different patterns were observed. The CTL group presented a higher number of failures (n = 5) at the distal connector. For the GZG group, the main failure mode originated at the occlusal surface of the pontic (n=6), and the SSG group presented the failure origin of seven samples located at the mesial connector. Fig. 5 presents the different failure patterns observed in each experimental group.

Fracture mapping of a representative sample from each experimental group is shown in Fig. 6.

4. Discussion

To our knowledge, this is the first study to test full-contour zirconia 3-unit fixed dental prostheses made of graded 3Y-TZP. According to the findings, the glass/silica infiltration techniques in the monolithic zirconia bridges significantly

increased the fatigue limits compared with the glazed control group (CTL). Therefore, the null hypothesis was rejected.

Glass- and silica-infiltrated zirconia ceramics were tested in previous studies in different geometric forms: plates, beams, discs and single crowns [6,21,27]. The specimen geometry may influence stress concentration [37], and the use of geometric forms similar to those used in the mouth for *in vitro* studies is important for its clinical relevance and prediction [6,38]. It has been reported that the fracture loads were lower in curved surfaces of anatomically correct crowns than in flat specimens [39].

The effects of minor additions of silica to 3Y-TZP have been discussed in previous studies that reported increase in resistance to low-temperature degradation (LTD), with no effect on resistance to slow crack growth (SCG). In the same way, mechanical properties, grain size, and densities were not affected [40,41]. In the present study, in the SSG group, the grooves generated by the CAD/CAM process were filled by silica. During the sintering process, silica on the zirconia surface can suffer a reaction in its solid state, resulting in zirconia sili-

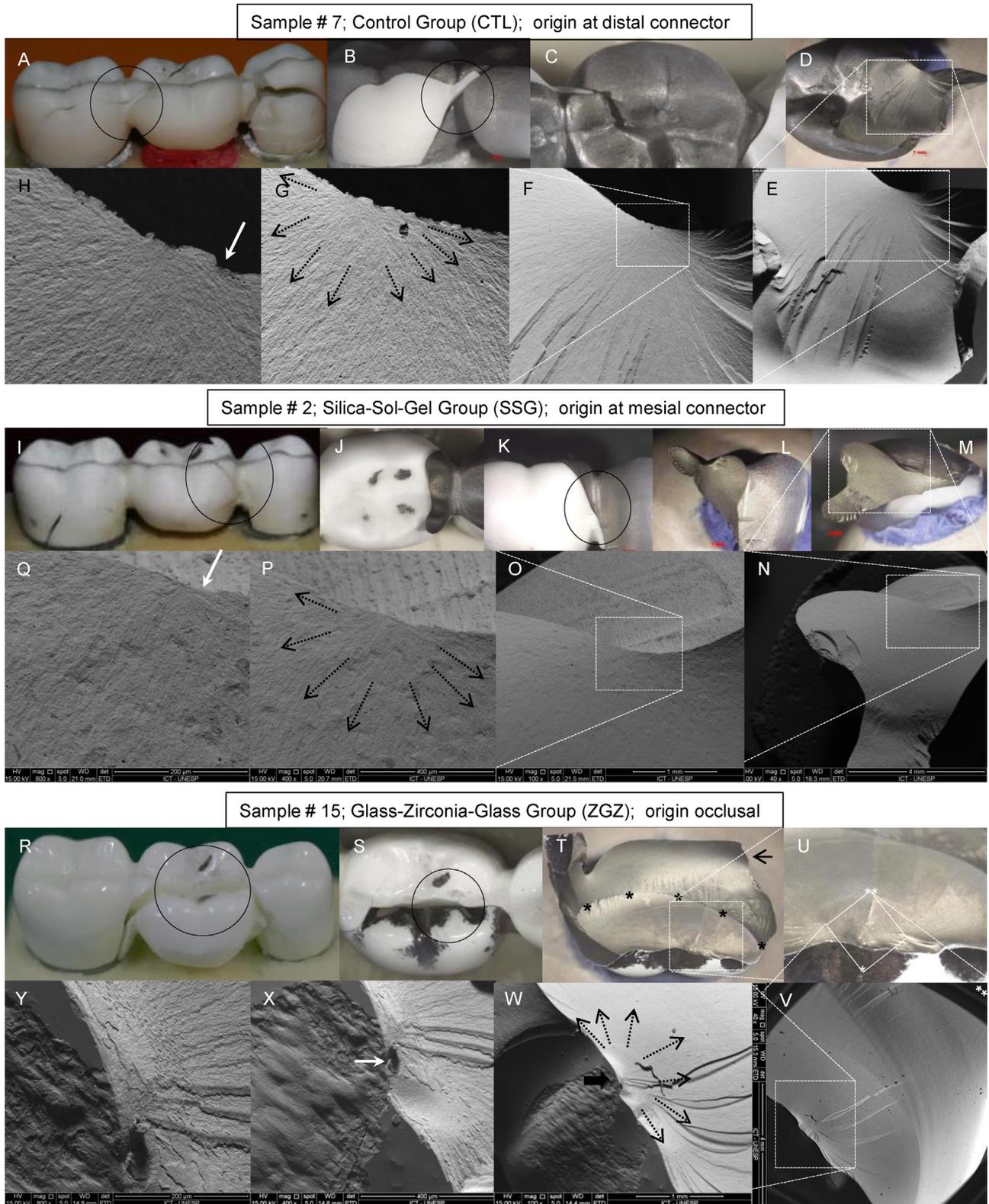


Fig. 6 – Fracture origin mapping from one sample of each experimental group. The black circles delimits the investigated areas. The dotted white lines mark the regions suspected of containing the fracture origin. The black arrows indicate the direction of crack propagation from the fracture initiator defect and the white arrows indicate the origin of the fracture. (A–H—clockwise): sample #7 from the control group (CTL): (A) the reassembly of the sample fragments indicates that the fracture ran from distal to mesial origin at the distal connector, (B) buccal view of the distal connector; (C) occlusal view of the pontic; (D) internal view of the distal connector; (E–H) SEM images of the suspected areas with crescent magnifications

cate, without the amorphous glassy phase. Therefore, silica is not in its free form. The layer of $ZrSiO_4$ formed on the 3Y-TZP surface reduces the number of flaws [27], and consequently, the possibility of these defects becomes the initial crack point in the ceramic material, when subjected to mechanical stress.

The increase in fatigue limit in the GZG group may be associated with the glass-filling of intergranular spaces and also by the residual glass layer on the surface of the infiltrated FDP, which has lower elastic modulus and better distributes stresses [42]. The higher variability in the results of GZG is probably due to variations in the material [43], such as different thicknesses of the residual glass layer.

Different load levels were used as initial loading for the fatigue test, depending on the group. This means, although the three groups are essentially the same in the bulk, their surface characteristics determined the resistance to monotonic loading and, finally, their resistance to fatigue.

The increase in structural homogeneity of silica infiltrated zirconia through sol-gel method had already been shown [27] which was confirmed by the low variability in the fatigue limit results found in the present study. We suggest that the silica infiltrated in the grooves produced by the milling process [27] and a lower tendency of zirconia for suffering degradation in water [44] explain the differences in the fatigue behavior of SSG and CTL specimens. Finally, the GZG group presented superior performance than monolithic zirconia and porcelain fused to zirconia specimens in several studies. Zhang and Kim [45] observed that glass-infiltrated zirconia showed better resistance to sliding damage than did monolithic zirconia and glazed zirconia, which can justify the higher fatigue limit than in the other groups. The interfacial fracture energy of graded zirconia, which had been previously veneered was superior than that of the non-graded counterparts [46].

For the silica infiltrated specimens and the other glazed specimens, failure was caused predominantly by flexural damage, where cracks grew to only a relatively small critical size and then quickly become unstable.

Chipping was not observed in this study, probably because chipping is most likely to occur in veneering porcelain under lower occlusal loads [3]. All failures were catastrophic, indicating a high resistance to fracture load. This is in agreement with Srikanth et al. [46], who observed an increase in flexural strength in glass-infiltrated specimens.

Unlike ductile materials, a crack does not initiate naturally in brittle ceramic materials. Typically, it is initiated from any pre-existing defects, and when in tension, they cause localized stress concentrations when loaded at more than a bearable level [47,48]. The analysis of the samples before sintering showed irregularities introduced from the CAM process, at the

mesial connector, and at the mesial surface of the premolar. We did not observe fractures departing from these sites in the control group, but we found origins away from these damage zones such as in the SSG specimens, which were mostly in the distal connector.

It is possible that an increase in the monoclinic phase at the cervical region [49] was not enough to initiate the fracture, and the major factor was the association between pre-existing flaws in the radii of the approximal curvature of the finishing line, as shown in the study by Oilo et al. [50]. But other factors may also play a role in the fractures of posterior fixed dental all-ceramic prostheses, including prosthesis design, flaw distribution, and direction and magnitude of occlusal loading [51].

In the current study, we attempt to simulate a clinical situation. Thus, we selected the worst scenario, according Kou et al. [52], with samples positioned 30° under loading on the buccal cusps. Furthermore, crown-shaped test specimens cemented on preparations of dentin-like material may bring the test conditions even closer to the clinical situation [53]. In addition, a wet environment could decrease the fracture resistance of ceramics [12,54,55]. However, despite the efforts to simulate a clinical scenario, the applied loads were not similar to the ones that naturally occur in the mouth (sliding contacts): and the tip contacted a portion of the non-functional cusps. Despite the limitations regarding the loading conditions, the fractographic analysis (macro and micro) showed that the fractures originated at occlusal surface always initiated in functional cusps, indicating the efficiency of methodology and results obtained.

5. Conclusions

Silica infiltration and glass infiltration were efficient methods for obtaining graded zirconia, since they improved the fatigue limit of the FDPs compared with glazed monolithic zirconia restorations.

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(40×; 100×; 400× and 1000× respectively). (L–Q—clockwise): sample #2 from the silica sol-gel group (SSG): (L) the reassembly of the sample fragments indicates that the fracture ran from mesial to distal; (J) occlusal view of the pontic; (K, L) buccal view of the mesial connector; (M) internal view of the mesial connector; (N–Q) SEM images of the suspected areas with crescent magnifications (40×; 100×; 400× and 800× respectively). (R–Y clockwise): sample #15 from the GZG group: (R) the reassembly of the sample fragments indicates that the fracture initiated from occlusal, ran to distal and then to mesial; (S) occlusal view of the pontic; (T) internal view of the buccal fragment. Black asterisks marks the arrest line, and the black arrow indicates the compression curl; (V–Y) SEM images of the suspected areas with crescent magnifications (40×; 100×; 400× and 800× respectively).

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